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## Application of Universal Scaled Reduced Temperature Parameter to the Three-Arm Star Polystyrene

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: (PS,  $M_w = 2.80 \times 10^5$ ,  $2.49 \times 10^6$  g/mol) t-  
 decalin 20 70 ,  
 $R_{G,Br,o}$   $(NR_{G,Br,o})^{2,3/2} t/t_c$   
 $(NR_{G,Li,o})^{2,3/2} t/t_c$  가  
 $t/t_c = [(T - Q_{\theta c}) / Q_{\theta c}] / [(Q_{\theta c} - T_c) / T_c]$  가  
 $N$  ,  $Q_{\theta c}$

**ABSTRACT** : Various chain sizes of 3-arm star polystyrenes (PS,  $M_w = 2.80 \times 10^5$ ,  $2.49 \times 10^6$  g/mol) in t-decalin solution were measured at the temperature range of 20 70 by means of viscometry and laser light scattering. In order to show universality in the expansion factor of 3-arm star polymer, it was expected that  $(NR_{G,Br,o})^{2,3/2} t/t_c$  would be used as an universal parameter, where  $R_{G,Br,o}$  was the unperturbed radius of gyration of star PS. However, much better universality had been observed when  $(NR_{G,Li,o})^{2,3/2} t/t_c$  parameter of the linear PS was used even for the 3-arm star PS. It could be explained if branching effect had been already taken into account in the part of  $t/t_c (= [(T - Q_{\theta c}) / Q_{\theta c}] / [(Q_{\theta c} - T_c) / T_c])$ . Here  $N$  and  $Q_{\theta c}$  stand for the number of monomer unit in a single polymer chain and a kind of theta temperature as the critical solution temperature  $T_c$  of the infinite molecular weight, respectively.

**Keywords** : expansion factor, universality, star polymer, scaled reduced temperature parameter.

1.  $1.5$   
 /  $Q$   
 가 (universality)  
 Flory  $Q$   
 ,  $Q$  가  
 가 Gaussian  $Q$  (excluded  
 (hard sphere) volume effect) Flory type



$$C_m = 1.423 \times 10^{-24} (\mathbf{u}^2/V_1)(R_0^2/M)^{-3/2}, \quad b = (V_1/\mathbf{u})^{1/2}/\mathbf{y}$$

(1) 가

(4) Candau

$$\dot{a}^5 - \dot{a}^3 = 2C_m \Psi [(T - Q)/Q] M^{1/2} g^{-3/2} + C' g^{-3} / \dot{a}^3 \quad (4)$$

$$\dot{a}^5 - \dot{a}^3 \doteq 2.846 \times 10^{-24} (v_3/V_1)^{1/2} (M/R_0^2)^{3/2} (\tau/\tau_c) g^{3/2} \quad (\text{if } \mathbf{a} = 1) \quad (5)$$

$$g = \frac{3f-2}{f^2} \quad (\text{at the ideal state of a regular f-arm star polymer}) \quad (6)$$

(4) (1)

$$C' \mathbf{a}^3 \quad (\mathbf{a} = 1)$$

가

$$g (= R_{G,Br}^2 / R_{G,Li}^2)$$

(6)

가

$\mathbf{h}$ 가

$$C(\text{g/mL})$$

Huggins

(7)

Kraemer

(8)

$$(\mathbf{h}\mathbf{h}_0)/\mathbf{h}_0 C = \mathbf{h}_{\text{ed}} = [\mathbf{h}] + k_H [\mathbf{h}]^2 C \quad (7)$$

$$\ln(\mathbf{h}/\mathbf{h}_0)/C = \mathbf{h}_{\text{inh}} = [\mathbf{h}] - k_K [\mathbf{h}]^2 C \quad (8)$$

$\mathbf{h}$

$[\mathbf{h}]$

$\mathbf{h}_{\text{ed}}$

$\mathbf{h}_{\text{inh}}$  inherent viscosity,  $k_H$   $k_K$  Huggins

Kraemer

$$R_v (= (3M[\mathbf{h}]/10p N_A)^{1/3})$$

$N_A$

가

(C = 0)

$\mathbf{q}$

(9)가

$R_{vv}$ 가

22 (9)

$P(\ )$

$(R_G)$

$$\frac{HC}{R_{vv}} = \frac{1}{M_w P(\dot{\epsilon})} \quad (\text{at } C \rightarrow 0)$$

$$\doteq \frac{1}{M_w} \left(1 + \frac{K^2 R_G^2}{3}\right) \quad (\text{at } KR_G < 1) \quad (9)$$

$$H = 4\delta^2 n_0^2 (dn/dc)^2 / (N_A \lambda_0^4), \quad K = 4\delta n_0 \sin(\dot{\epsilon}/2) / \lambda_0$$

$H$

$R_{vv}$

$K$

$(\lambda_0)$

Rayleigh

$C$

(g/mL)

$n_0$

$M_w$

Flory

$P(x)$

$$\text{Debye } [P(x) = (2/x^2)(e^{-x} + x - 1), \quad x = (KR_G)^2]$$

(9)

가

$(KR_G < 2)$

$R_G$

$I(t)$

(time correlator)

(time correlation function)

$$|g^{(1)}(t)|$$

$z$ -

cumulant

$$|g^{(1)}(t)|$$

(10)

$$\ln |g^{(1)}(t)|$$

$t$

23

$$\ln |g^{(1)}(t)| = -\langle \dots \rangle t + (1/2!) \langle \dots \rangle^2 t^2 + \dots \quad (10)$$

$$\langle \dots \rangle / K^2 = D_0 (1 + k_g K^2 R_G^2) (1 + k_D C) \quad (11)$$

$$\ln |g^{(1)}(t)|$$

$K$

$C$

$z$ -

$\langle \dots \rangle$

$D_0$

(11)

$\mathbf{m}$

2

$k_g$

$k_D$

2

$(KR_G/1)$   
 $D_o < > /K^2$   
 Einstein (12)

$(C/1)$   
 $D_o \gamma$

$(11)$   
 Stokes-  
 $R_H$

$2.49 \times 10^6$  g/mol  
 $M_w/M_n = 1.30$   
 $0.28$  M (=  $2.80 \times 10^5$  g/mol)

$1.07$   
 $2.49$  M  
 $0.28$  M

Shultz (13)  $\gamma$   
 $R_G$   $R_H$

Table 1

$R_H = k_B T / 6\pi \eta D_o$  (Stokes-Einstein) (12)

$k_B$   $h$  Boltzmann

$w(M)dM = \lambda^{Z+1} + 1 M^Z e^{-\lambda M} dM / (Z+1)$  (13)

$\lambda = Z/M_n = (Z+1)/M_w = (Z+2)/M_z$  (14)

$R_w^2 = R_z^2 (Z+1)/(Z+2)$  (15)

$(x)$  gamma,  $R_w$   $R_z$   
 $z$ - 2.49 M  $Z$   
 variance (=  $\sigma^2$ )  $M_z/M_w$   
 $Z=3.8$

$M_w > 10^6$  g/mol 가  
 $(T_c)$

PS  
 $R_G^2/M_w = 9.4 \times 10^{-18}$  cm<sup>2</sup>/(g/mol) 2.49 M  
 $g = 0.87 [(45.1/48.4)^2]$   
 $g = 0.78$

PS trans-decahydronaphthalin (t-decalin) 가  
 PS Polymer Sources PS/t-Decalin  $t/t_c$

PS  $T_c$

PS  $T_c$

가 Table 1

PS 2 /hour  
 benzene/cyclohexane (w/w = 1/4) (automatic recording turbidimeter)  
 ( 1 wt%) 20 mL  $(T_p) \pm 0.02$   
 nonsolvent methanol ( 0.2 mL/hour) Figure 1 PS/t-decalin  $T_c$   
 가 2.49 M (=  $1/T_c$   $1/M_w^{1/2}$ ) Figure 2

**Table 1. Characteristics of 3-Arm Star Polystyrene Samples**

$M_w$ ( $10^6$ g/mol)	$M_w/M_n$	$R_{G,Br,o}^*$ (nm)	$R_{H,Br,o}^*$ (nm)	$[\eta]^*$ (mL/g)	$R_v$ (nm)	$R_G/R_H$	$R_v/R_H$
0.28	1.07	-	11.7	44.1	12.5	-	1.07
2.49	1.30	47.5	37.2	137.6	37.9	1.28	1.02
	(1.07)**	(45.1)**	(35.3)**			(1.28)**	(1.07)**

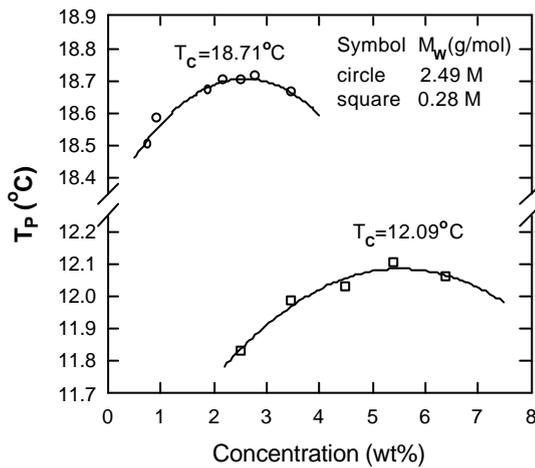
\* The unperturbed values measured at  $T_c$  temperature (= 22.2 °C) in t-decalin.  
 \*\* The expected values on the assumption of the same polydispersity as the 0.28 M sample.

Figure 2

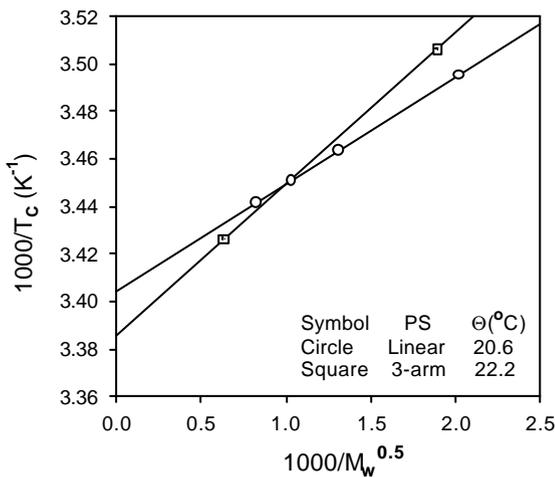
PS 20.6 PS 1.6

$T_c = 22.2 \pm 0.2$

$$\frac{1}{T_c} = \frac{1}{295.37} + \frac{0.06370}{\sqrt{M_w}} \quad (3\text{-arm star PS/t-decalin}) \quad (16)$$

$$C_c (\text{wt}\%) = 8.5 \times 10^2 M_w^{-0.40} \quad (3\text{-arm star PS/t-decalin}) \quad (17)$$


**Figure 1.** Concentration dependence of the phase transition temperature  $T_P$  in t-decalin solution of two 3-arm PS samples. The peak point of  $T_P$  curve was considered as a critical point.



**Figure 2.** Plots of  $1/T_c$  vs  $1/M_w^{1/2}$  in the linear and 3-arm star PS/t-decalin.

$-0.40 \pm 0.05$  25.26

$-0.38$  가

2.49 M 2  $A_2 = 0$

Flory  $A_2$  Figure 3  $T_c (=$

$23.0 \pm 0.3$  가  $21$  가

$22.2$  )  $A_2$  가  $0.8$  가

Benoit Flory (  $A_2$  )

$a = 1$   $\alpha$

$\alpha \approx T_c$  ,

g/mL inherent viscosity 가  $0.7 \times 10^{-3}$

mer (BI9000AT)가 Krae-

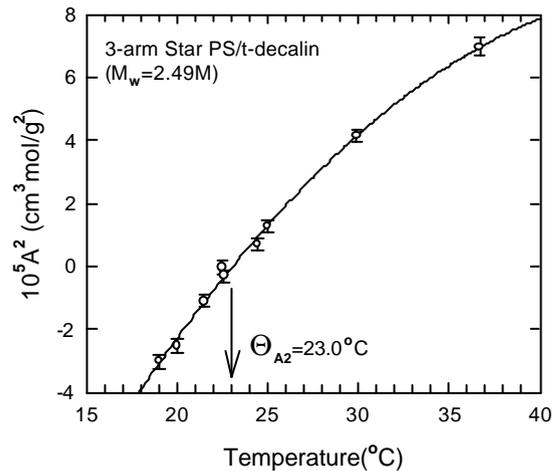
haven (Lexel model 95) Brook-

nm가 Ar (Lexel model 95) 514

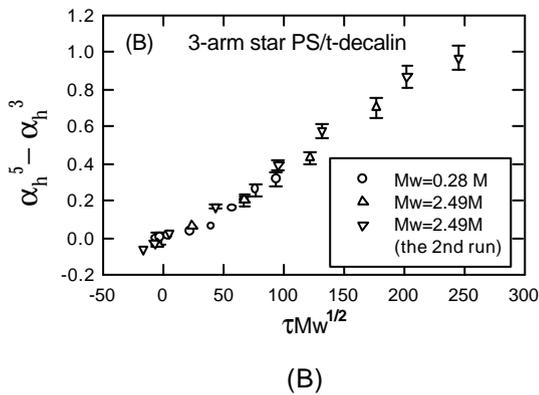
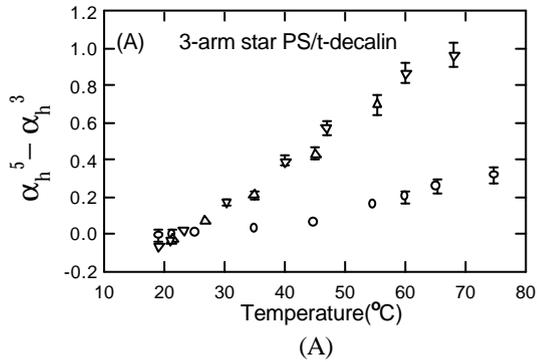
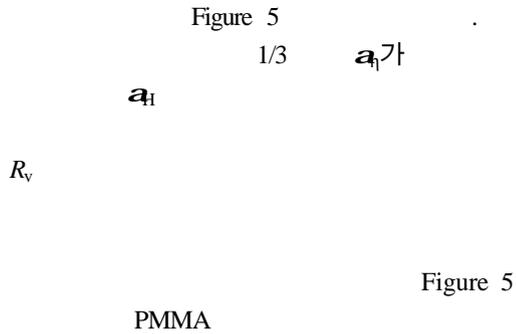
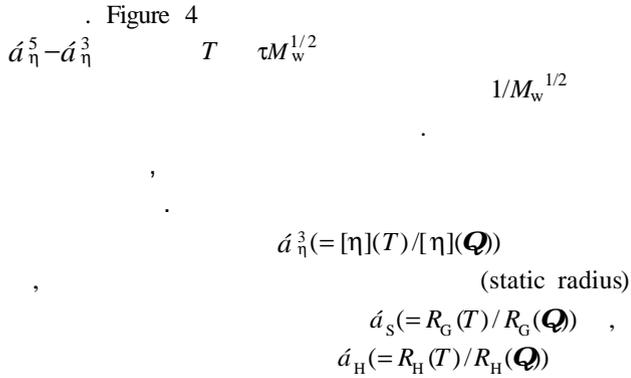
(  $C = 2 \times 10^{-4}$  g/mL )

4.

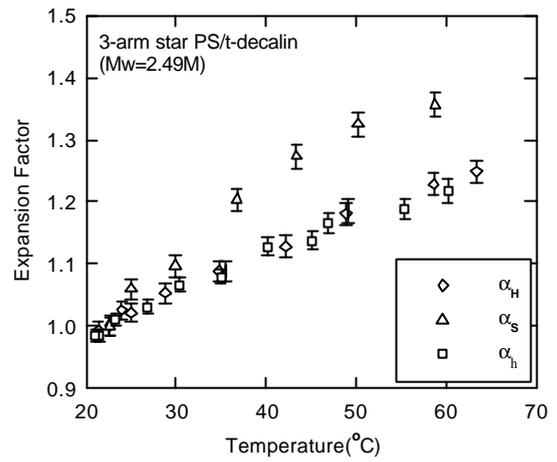
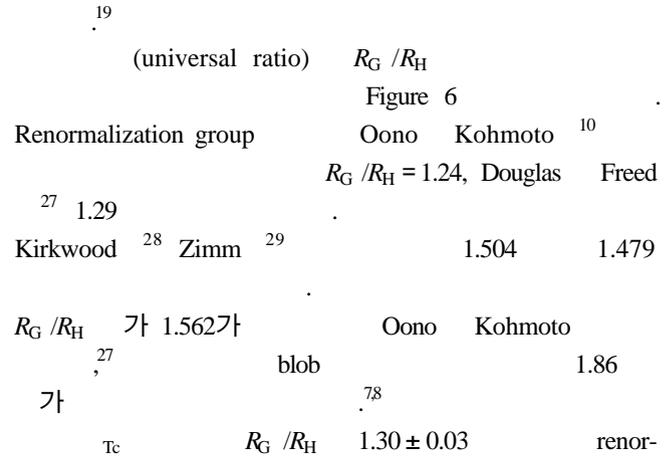
PS ( $M_w = 0.28 \text{ M}, 2.49 \text{ M}$ )



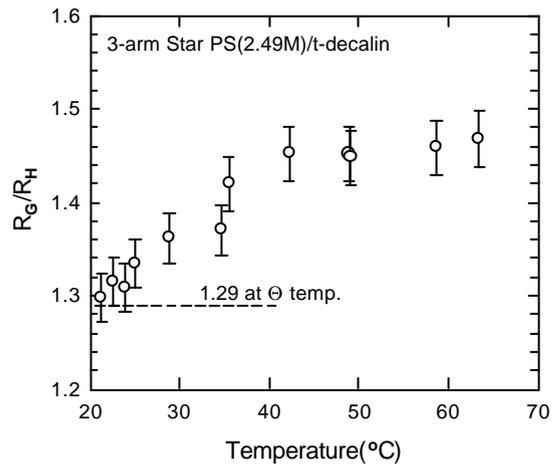
**Figure 3.** Plot of the second virial coefficient,  $A_2$  as a function of temperature in a 3-arm star PS (2.49 M)/t-decalin.



**Figure 4** Plots of  $\dot{a}_\eta^5 - \dot{a}_\eta^3$  of intrinsic viscosity as a function of temperature (A) and  $\tau M_w^{1/2}$  parameter (B) in the system of 3-arm star PS/t-decalin.

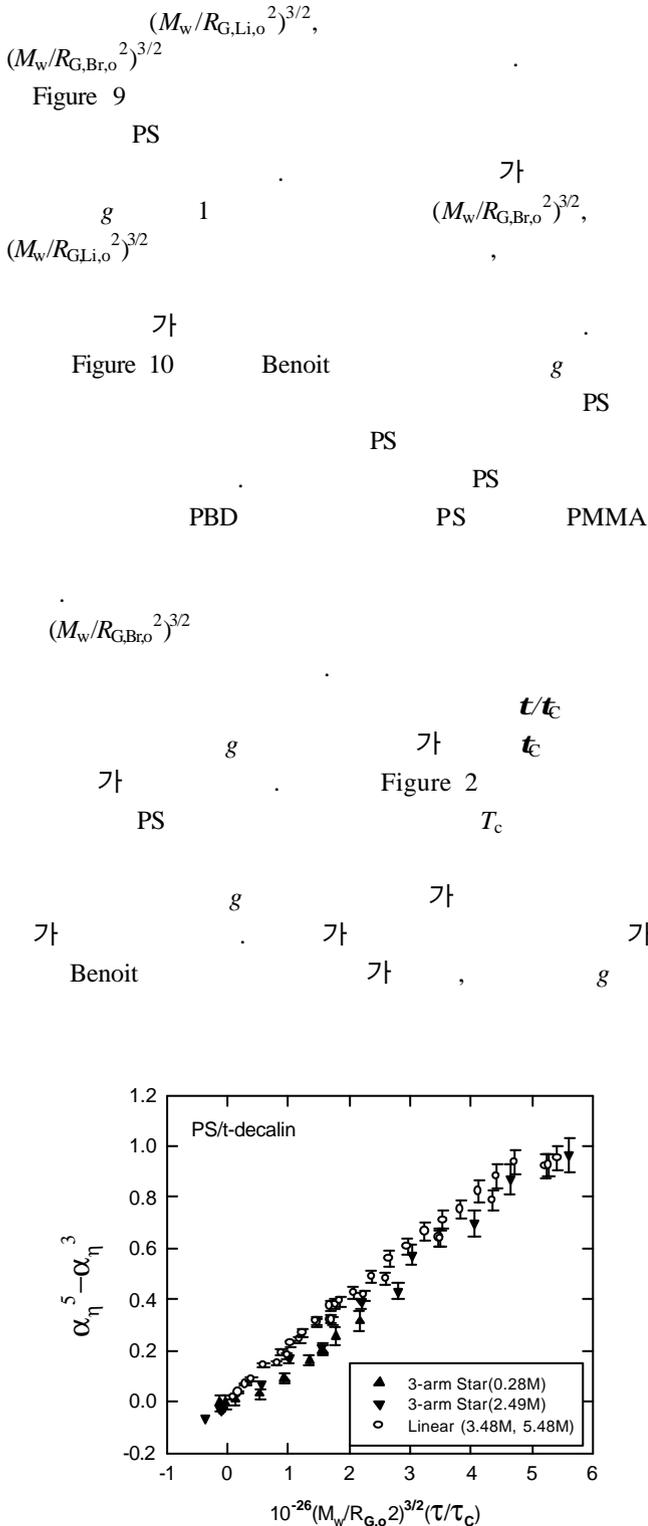


**Figure 5.** Plots of three kinds of expansion factors,  $a_H$ ,  $a_s$  and  $a_h$ , as a function of temperature in the system of 3-arm star PS (2.49M)/t-decalin.

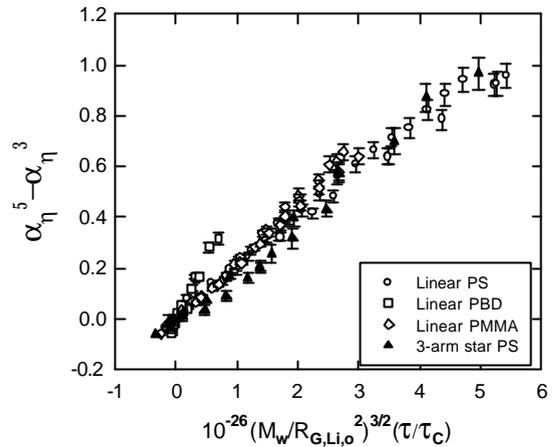


**Figure 6.** Plot of universal ratio of  $R_G/R_H$  as a function of temperature in the system of 3-arm star PS (2.49 M)/t-decalin.





**Figure 9.** Plots of  $\alpha_5 - \alpha_3$  as a function of  $(M_w/R_{G,0}^2)^{3/2} t/t_c$  in the systems of 3-arm star and linear PS/t-decalin. Here  $M_w/R_{G,Br,0}^2$  and  $M_w/R_{G,Li,0}^2$  were used as their scaling constants for star PS polymers and for linear PS samples, respectively.



**Figure 10.** Plots of  $\alpha_5 - \alpha_3$  as a function of  $(M_w/R_{G,Li,0}^2)^{3/2} t/t_c$  in the systems of linear PS, linear PBD, linear PMMA, and 3-arm star PS. Here the scaling constant of linear PS polymer was used for star PS polymers and their scaling constants were used for other linear polymer samples.

Figure 2

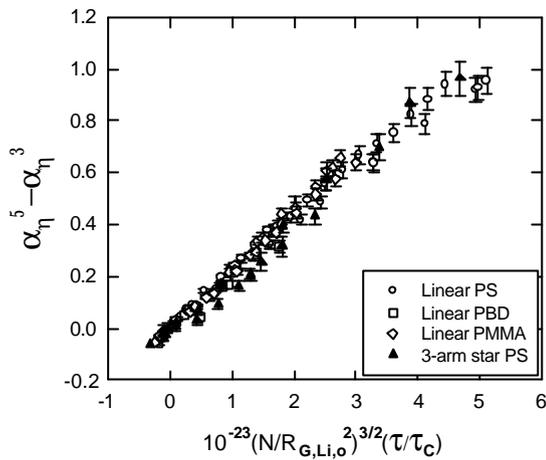
Figure 10

Figure 11

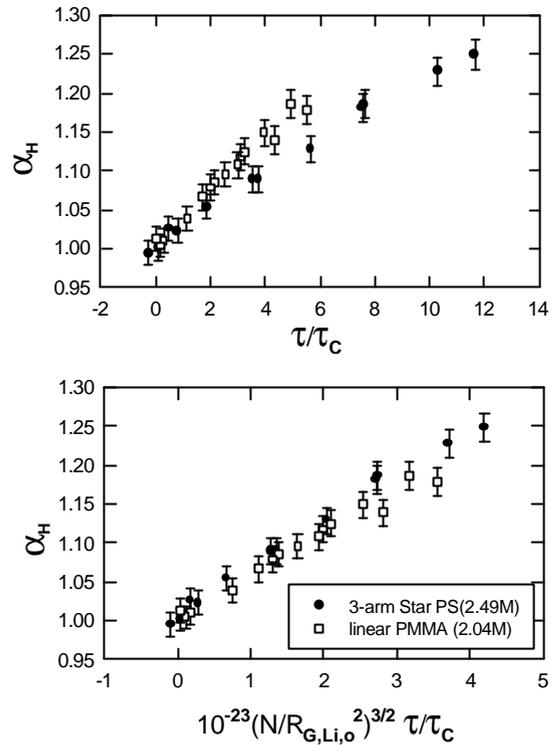
Figure 12

Figure 8

Figure 11



**Figure 11.** Plots of  $\alpha_{\eta}^5 - \alpha_{\eta}^3$  as a function of  $(N/R_{G, Li, o}^2)^{3/2} \tau/\tau_c$  in the systems of linear PS, linear PBD, linear PMMA, and 3-arm star PS. Here all calculation in  $(N/R_{G, Li, o}^2)^{3/2} \tau/\tau_c$  was the same as  $(M_w/R_{G, Li, o}^2)^{3/2} \tau/\tau_c$  parameter except substitution of molecular weight  $M_w$  with the number of monomer in a single chain  $N$ .



**Figure 12.** Plots of the expansion factor of hydrodynamic radius,  $\alpha_H$  as a function of  $\tau/\tau_c$  and  $(N/R_{G, Li, o}^2)^{3/2} \tau/\tau_c$  in the systems of 3-arm star PS and linear PMMA.

가

Figure 8 Figure 11

5.

1) 가  
 $\tau M_w^{1/2}$   
 $(N/R_{G, Bro}^2)^{3/2} \tau/\tau_c$  가  
 $(N/R_{G, Lr, o}^2)^{3/2} \tau/\tau_c$  가  
 2) 가  
 $\tau/\tau_c$  가  
 가  
 가

3)

PS/t-decalin  
 $\alpha_H$   
 PMMA/n-butyl chloride  
 $\alpha_H$   
 $\alpha_{\zeta}^3 = \alpha_{\zeta}^2 \alpha_H$   
 Weill  
 20  
 30  
 : 2001

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